

# Degassing phenomena during sintering and crystallization of glass powders

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## Background

Degassing may cause trouble in manufacture of sintered glass-ceramics via the formation of bubbles (deteriorate mechanical strength or surface quality).

## Experimental

Alkali-silicate-glass powders ( $D \approx 1 \text{ mm}$  and  $30 \mu\text{m}$ ) were studied by MS-analysis of evolved gases at  $1\text{E-}3 \text{ mbar}$  during heating ( $40 \text{ K/min}$ ), and by DTA and dilatometry.

## Results

Complex degassing behaviour is caused by various mechanism (Fig.1). Examples are (see paper for more detail):

- b** Diffusive gas transport from the bulk (gas release is more pronounced and may be even almost fully exhausted at low temperature for fine powders).
- d** Sintering strongly decreases degassing (diffusion length).
- e** Primary crystallization causes a sharp degassing peak (structural changes can promote transport of volatiles).
- f** Secondary crystallization causes a gas release peak for coarse powders (degassing of fine powders is already exhausted).
- g** Melting of crystalline phases can decrease water release (may be due to a lower water solubility in the former crystal volume fraction).
- h** Bursting bubbles accelerate degassing after melting of the crystal phases (if gases are still soluted in the melt).

## Summary

Degassing of crystallizing glass powders is sensitive to various phenomena (bulk diffusion, sintering, crystallization, melting). Evolved gas analysis backed up by DTA and shrinkage measurements is a powerful tool for optimizing sinterability of glass powders.

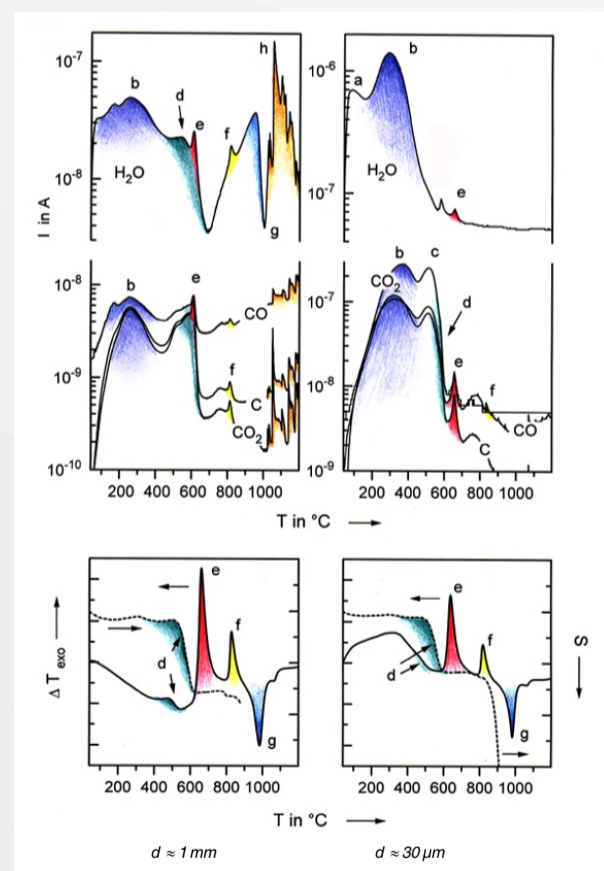


Fig. 1  
Top: Dynamic degassing for coarse (left) and fine (right) glass powder (upper curve  $\text{H}_2\text{O}$ ; middle  $\text{CO}_2$ ,  $\text{CO}$ ,  $\text{C}$ ). Bottom: DTA curves (left ordinates) and linear shrinkage (right ordinates) in arb. units.